ISSN 1674-1056
http://cpb.iphy.ac.cn
http://iopscience.iop.org/cpb
中国物理 B
Chinese Physics
中国物理 Chinese Physics B

Direct deposition of graphene nanowalls on the ceramic powders for the fabrication of ceramic matrix composite

Hai-Tao Zhou(周海涛), Da-Bo Liu(刘大博), Fei Luo(罗飞), Ye Tian(田野), Dong-Sheng Chen(陈冬生), Bing-Wei Luo(罗炳威), Zhang Zhou(周璋), Cheng-Min Shen(申承民)

Citation: Chin. Phys. B . 2019, 28(6): 068102. **doi:** 10.1088/1674-1056/28/6/068102
Journal homepage: http://cpb.iphy.ac.cn; http://iopscience.iop.org/cpb

What follows is a list of articles you may be interested in

Superlubricity enabled dry transfer of non-encapsulated graphene

Zhe Ying(应哲), Aolin Deng(邓奥林), Bosai Lyu(吕博赛), Lele Wang(王乐乐), Takashi Taniguchi, Kenji Watanabe, Zhiwen Shi(史志文)

Chin. Phys. B . 2019, 28(2): 028102. **doi:** 10.1088/1674-1056/28/2/028102

Electronic, optical property and carrier mobility of graphene, black phosphorus, and molybdenum disulfide based on the first principles

Congcong Wang(王聪聪), Xuesheng Liu(刘学胜), Zhiyong Wang(王智勇), Ming Zhao(赵明), Huan He(何欢), Jiyue Zou(邹吉跃)

Chin. Phys. B . 2018, 27(11): 118106. **doi:** 10.1088/1674-1056/27/11/118106

Dynamically tunable terahertz passband filter based on metamaterials integrated with a graphene middle layer

MaoSheng Yang(杨茂生), LanJu Liang(梁兰菊), DeQuan Wei(韦德泉), Zhang Zhang(张璋), Xin Yan(闫昕), Meng Wang(王猛), JianQuan Yao(姚建铨)

Chin. Phys. B . 2018, 27(9): 098101. **doi:** 10.1088/1674-1056/27/9/098101

Novel graphene enhancement nanolaser based on hybrid plasmonic waveguides at optical communication wavelength

Zhengjie Xu(徐政杰), Jun Zhu(朱君), Wenju Xu(徐汶菊), Deli Fu(傅得立), Cong Hu(胡聪), Frank Jiang

Chin. Phys. B . 2018, 27(8): 088104. **doi:** 10.1088/1674-1056/27/8/088104

Modulated thermal transport for flexural and in-plane phonons in double-stub graphene nanoribbons

Chang-Ning Pan(潘长宁), Meng-Qiu Long(龙孟秋), Jun He(何军)

Chin. Phys. B . 2018, 27(8): 088101. **doi:** 10.1088/1674-1056/27/8/088101

ISSN 1674-1056

CPB
Chinese Physics B

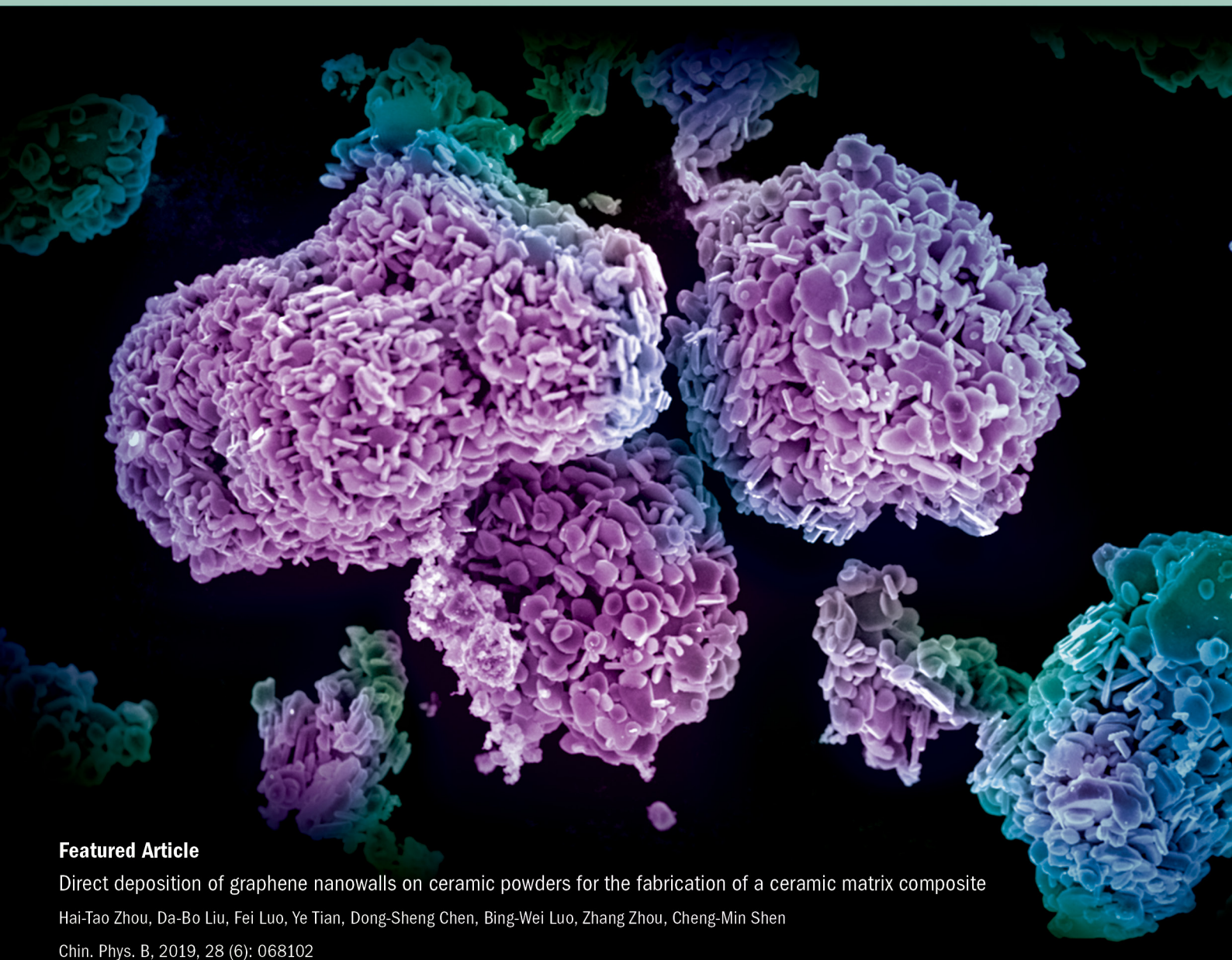
Volume 28 June 2019 Number 6

TOPICAL REVIEW

- Topological superconductors

A series Journal of the Chinese Physical Society Distributed by IOP Publishing

iopscience.org/cpb | **cpb.iphy.ac.cn**



Featured Article

Direct deposition of graphene nanowalls on ceramic powders for the fabrication of a ceramic matrix composite

Hai-Tao Zhou, Da-Bo Liu, Fei Luo, Ye Tian, Dong-Sheng Chen, Bing-Wei Luo, Zhang Zhou, Cheng-Min Shen

Chin. Phys. B, 2019, 28 (6): 068102

CHINESE PHYSICAL SOCIETY | IOP Publishing

Chinese Physics B (中国物理 B)

Published monthly in hard copy by the Chinese Physical Society and online by IOP Publishing, Temple Circus, Temple Way, Bristol BS1 6HG, UK

Institutional subscription information: 2019 volume

For all countries, except the United States, Canada and Central and South America, the subscription rate per annual volume is UK£974 (electronic only) or UK£1063 (print + electronic).

Delivery is by air-speeded mail from the United Kingdom.

Orders to:

Journals Subscription Fulfilment, IOP Publishing, Temple Circus, Temple Way, Bristol BS1 6HG, UK

For the United States, Canada and Central and South America, the subscription rate per annual volume is US$1925 (electronic only) or US$2100 (print + electronic). Delivery is by transatlantic airfreight and onward mailing.

Orders to: IOP Publishing, P. O. Box 320, Congers, NY 10920-0320, USA

Supported by the China Association for Science and Technology and Chinese Academy of Sciences

Editorial Office: Institute of Physics, Chinese Academy of Sciences, P. O. Box 603, Beijing 100190, China

Tel: (86 - 10) 82649026 or 82649519, Fax: (86 - 10) 82649027, E-mail: cpb@aphy.iphy.ac.cn

主管单位: 中国科学院
主办单位: 中国物理学会和中国科学院物理研究所
主　　编: 欧阳钟灿
出　　版: 中国物理学会
印刷装订: 北京科信印刷有限公司
编　　辑: Chinese Physics B 编辑部
国内发行: Chinese Physics B 出版发行部
国外发行: IOP Publishing Ltd
发行范围: 公开发行

国际统一刊号: ISSN 1674–1056
国内统一刊号: CN 11–5639/O4
编辑部地址: 北京 中关村 中国科学院物理研究所内
通 讯 地 址: 100190 北京 603 信箱
电　　话: (010) 82649026, 82649519
传　　真: (010) 82649027
"Chinese Physics B"网址:
http://cpb.iphy.ac.cn
http://iopscience.iop.org/journal/1674-1056

Published by the Chinese Physical Society

RAPID COMMUNICATION

Direct deposition of graphene nanowalls on ceramic powders for the fabrication of a ceramic matrix composite*

Hai-Tao Zhou(周海涛)[1], Da-Bo Liu(刘大博)[1], Fei Luo(罗飞)[1], Ye Tian(田野)[1], Dong-Sheng Chen(陈冬生)[1], Bing-Wei Luo(罗炳威)[1,†], Zhang Zhou(周璋)[2], and Cheng-Min Shen(申承民)[2,‡]

[1] *Beijing Institute of Aeronautical Materials, Aero Engine Corporation of China, Beijing* 100095, *China*

[2] *Beijing National Laboratory of Condensed Matter Physics, Institute of Physics, Chinese Academy of Sciences, Beijing* 100190, *China*

(Received 28 March 2019; revised manuscript received 3 April 2019; published online 8 May 2019)

Uniform mixing of ceramic powder and graphene is of great importance for producing ceramic matrix composite. In this study, graphene nanowalls (GNWs) are directly deposited on the surface of Al_2O_3 and Si_3N_4 powders using chemical vapor deposition system to realize the uniform mixing. The morphology and the initial stage of the growth process are investigated. It is found that the graphitic base layer is initially formed parallel to the powder surface and is followed by the growth of graphene nanowalls perpendicular to the surface. Moreover, the lateral length of the graphene sheet could be well controlled by tuning the growth temperature. GNWs/Al_2O_3 powder is consolidated by using sparking plasma sintering method and several physical properties are measured. Owing to the addition of GNWs, the electrical conductivity of the bulk alumina is significantly increased.

Keywords: graphene, ceramic matrix composite, chemical vapor deposition

PACS: 81.05.ue, 81.05.Mh, 81.15.Gh **DOI:** 10.1088/1674-1056/28/6/068102

1. Introduction

Monolithic ceramics are widely used in the materials industry, including wear-resistance parts, electronics, and various coatings. However, their applications are significantly limited by the brittleness and poor electrical conductance. To overcome these drawbacks, ceramic composites have been developed by incorporating nanofillers into the ceramic matrices.[1–6] Recently, graphene has been widely investigated as a good filler for producing highly tough and stiff ceramic matrix composite (CMC).[7–15] According to the papers reported in this field, the main challenges have been to produce high quality graphene and realize good dispersion of graphene in ceramic matrix. In these studies, ball-milling[7–12] and colloidal processing[13–15] were two mainly used mixing methods. Ball-milling technique imposed heavy shear forces on graphene to break its agglomeration and promote its dispersion. Colloidal processing produced well dispersed mixture usually by adding graphene oxide (GO) dropwise into ceramic suspension under magnetic stirring. However, the severe collision in ball-milling and the oxidation treatment in preparing GO by hummer's method would all significantly damage the graphene flakes and give rise to massive structural defects,[16,17] which degraded the physical properties in contrast with pure graphene. To produce high quality CMC, it is necessary to develop a new method to realize better control of the quality of graphene, as well as uniform mixing.

Recent years have witnessed great progress in the growth of graphene nanowalls (GNWs) on arbitrary substrates by employing plasma-enhanced chemical vapor deposition (PECVD) technique.[18–23] This method enables the low-temperature and uniform synthesis of graphene nanosheets owing to the presence of reactive species generated in the plasma region, and the GNWs have been widely used in the aspects of solar cells, supercapacitor, and field emission.[19,20,22] In this study, we aim to utilize PECVD to directly deposit graphene on the surface of ceramic powder, and the morphology and the initial stage of growth process of GNWs have been investigated.

2. Experimental procedure

2.1. Growth of GNWs on the ceramic powder

The α -alumina powder (Xilong Scientific Ltd.) with a purity of 99.85% and an average particle size of 50 μm was studied. GNWs were deposited in a remote radio-frequency (RF) PECVD system. Thin powder layer was flatly laid in the bottom of the crucible and then transferred into the quartz tube furnace. Prior to deposition, the furnace was pumped down to a pressure of 5×10^{-1} Pa, then the temperature was raised to 600 °C or 700 °C at a rate of 20 °C /min under the protection of Ar flow of 150 standard cubic centimeters per minute

*Project supported by the National Natural Science Foundation of China (Grant Nos. 51602300 and 51602299) and the National Key Research and Development Program of China (Grant No. 2018FYA0305800).

†Corresponding author. E-mail: luobingwei@126.com

‡Corresponding author. E-mail: cmshen@iphy.ac.cn

(sccm). Thereafter, CH_4 flow (20 sccm) was introduced into the furnace and 290 W RF was generated simultaneously. After the deposition, the RF and CH_4 flow were turned off and the ceramic powder was naturally cooled down to room temperature.

2.2. Spark plasma sintering (SPS)

Bulk ceramic composite was fabricated using an SPS-1050 T apparatus. The powder was loaded in a cylindrical graphite die with an inner diameter of 40 mm. Two graphite plungers were used to seal the die on both ends. The sintering process was carried out under a vacuum of 5 Pa and a uniaxial pressure of 4 MPa was applied. Samples were heated to 1450 °C at a rate of 10 °C/min. The dwelling time was 5 minutes.

2.3. Characterizations

The microstructure of the powder was characterized by scanning electron microscope (SEM, Hitachi S-4800) and high resolution transmission electron microscope (HRTEM, FEI Tecai G2F20). Raman spectrum was recorded on the Horiba Jobin Yvon LabRAM HR-800 with a laser wavelength of 532 nm and an incident power of 1 mW. The x-ray photoelectron spectroscopy (XPS) analysis was carried out on Thermo escalab 250XI, using monochromatized Al Kα radiation at 150 W. The high resolution XPS spectra were recorded in the constant analyzer energy (CAE) mode with a pass energy of 20 eV and a step size of 0.1 eV. The density values were measured using the Archimedes method. Vecker hardness tests were carried out on the Future-Tech with a 100-g force. Keysight 2902A was used to measure the conductivity at 25 °C on a small piece of sample with a size of 8 mm×8 mm×0.5 mm.

3. Results and discussion

Figure 1(a) shows a schematic illustration of the remote PECVD system, in which the RF helical-coil is mounted at the upstream side and could dissociate the methane gas into various active radicals. Owing to these radicals, the growth of GNWs does not depend on the catalytic decomposition of the precursor by the metallic substrate and could be realized on almost arbitrary substrates from metal to insulator, such as copper, silicon, and glass.[22,24,25] Figure 1(b) shows a comparison photograph, from which we could clearly find that the alumina (Al_2O_3) powder completely turns black after the GNWs deposition of 90 min. Raman spectrum in Fig. 1(c) verifies the

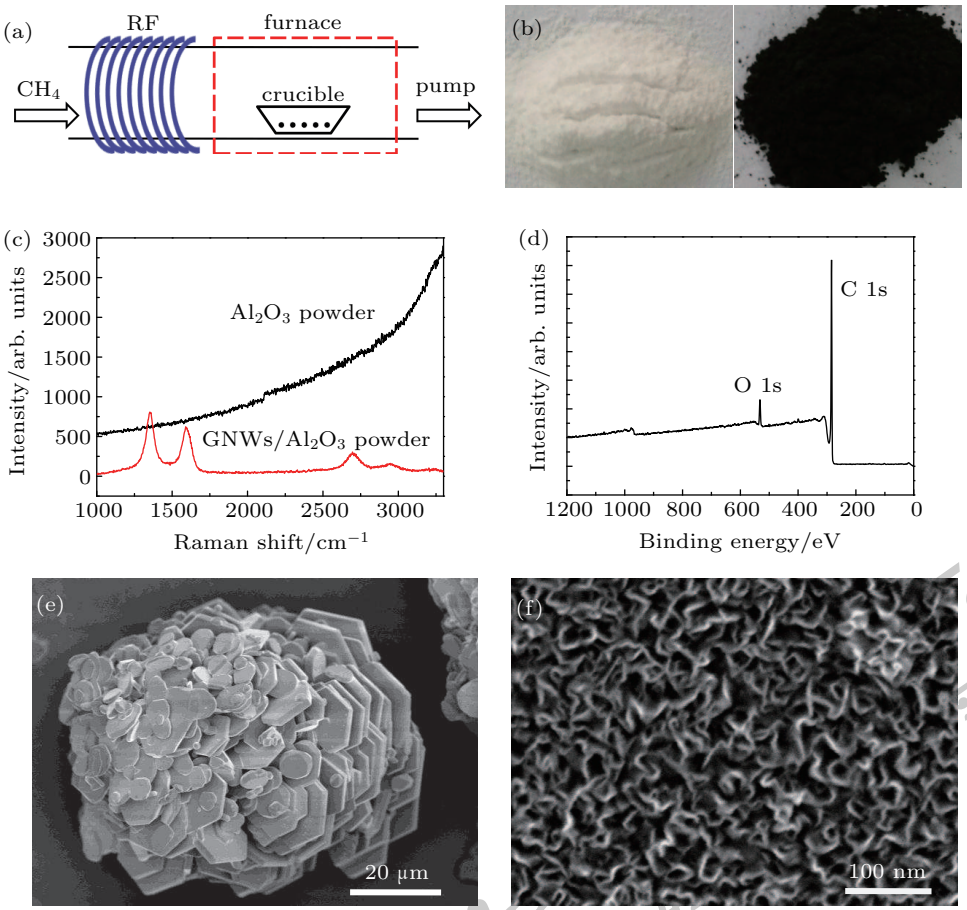


Fig. 1. (a) Schematic illustration of PECVD system. (b) Photographs of Al_2O_3 powder before (left) and after (right) GNWs deposition. (c) Raman spectrum. (d) XPS of GNWs/Al_2O_3 powder. (e) Low-magnification SEM image showing overall view of Al_2O_3 particles. (f) High-magnification image showing GNWs on the particle. Graphene deposition parameter: 700 °C, 290 W, 90 min.

formation of GNWs and presents three intrinsic peaks located at 1353 cm^{-1}, 1590 cm^{-1}, and 2697 cm^{-1}, corresponding to D band, G band, and 2D band of grapheme, respectively.[26,27] XPS analysis is used to study the chemical composition of the GNWs. Figure 1(d) is a representative spectrum, showing a strong C 1s peak at 284.8 eV and a small O 1s peak at 532.5 eV. No other elements are detected in the GNWs except for some oxygen adsorbates. The existence of these oxygen adsorbates is due to the physical and/or chemical adsorption of oxygen when the sample is exposed to ambient conditions.

The large-scale SEM image in Fig. 1(e) is an overall view of a single alumina particle. The particle has a size of ~ 50 μm and consists of numerous alumina pieces. The morphology and structure of the GNWs-capping on the alumina particle are analyzed by the SEM and HRTEM techniques. From the higher-magnification SEM image (Fig. 1(f)), GNWs could be clearly identified, showing a curled-sheet feature and an average length of ~ 70 nm. According to the previous report, GNWs are composed of two kinds of structures, named stem- and branch-structures.[23] The graphene sheets visible in Fig. 1(f) are referred to as stem. As shown in Fig. 2(a), we could see the branches at much higher magnification, in which several stem- and branch- structures are labeled by the solid and dashed arrows, respectively. The branches are much smaller and dimmer compared with the stems. To further understand the structure of GNWs,TEM is harnessed to study the morphology. Figure 2(b) shows a typical TEM image of a single sheet of GNWs. Dense branches with the size of several nanometers could be clearly identified, and three of them are labeled by arrows to guide the eye. HRTEM images could be used to identify the graphene layers. The image (Fig. 2(c)) obtained at the edge of stem-structure indicates that this stem is terminated with two-layer graphene. The interlayer spacing between two neighboring monolayer is about 0.37 nm. This value is slightly larger than that of graphite, which is most likely due to the reduced interactions between graphene layers.[19] Figure 2(d) shows an HRTEM image recorded in the zone of branches, from which we could find that the branches are composed of various graphene layers with an average interlayer spacing of 0.374 nm. These results imply that the GNWs films correspond to few-layer graphene.

Many efforts have been made to investigate the growth mechanism of GNWs on metallic substrates, such as Cu, Ni, and Pt.[22,23,28] Most of these studies indicated that the growth of GNWs was composed of two stages: initially, the carbon radicals nucleated and massive graphene layers in parallel with substrate took form, following a two-dimensional (2D) growth mode; as the number of layers increased and the strain energy accumulated, 2D layers became energetically unfavorable and a transition to three-dimensional (3D) growth took place. However, few work reported the growth mechanism on the surface of ceramics. To elucidate this issue, the growth process of GNWs is monitored by gradually varying the growth time under otherwise the same condition. Before graphene deposition, the surface of alumina particle has a clean and smooth topography, as shown in the SEM image of Fig. 3(a). After 10 min deposition (Fig. 3(b)), graphene sheets with wrinkle structures could be clearly identified. At this stage, these sheets are still lying down on the substrate. When the deposition time is elongated to 20 min (Fig. 3(c)), the transition to 3D growth occurs and some graphene sheets stand up at the substrate, as labeled by solid arrows. The above observations prove that the growth of GNWs on ceramic surface also complies well with the two-stage mode. According to this mode, we could control the lateral width of the graphene sheets by changing the growth temperature.[22] As shown in Figs. 3(d) and 3(e), at a growth temperature of 700 °C, all the sheets of GNWs present similar width of 70 nm; in contrast, the width increases to 150 nm at 600 °C. In ball-milling and colloidal methods, the size of graphene flakes is incontrollable. Thus, no work so far has reported the effect of the size distribution of graphene sheets on the properties of CMC. In our experiment, the size of graphene sheets is uniform at a given growth temperature and is adjustable by changing the temperature, which makes the study of size effect feasible.

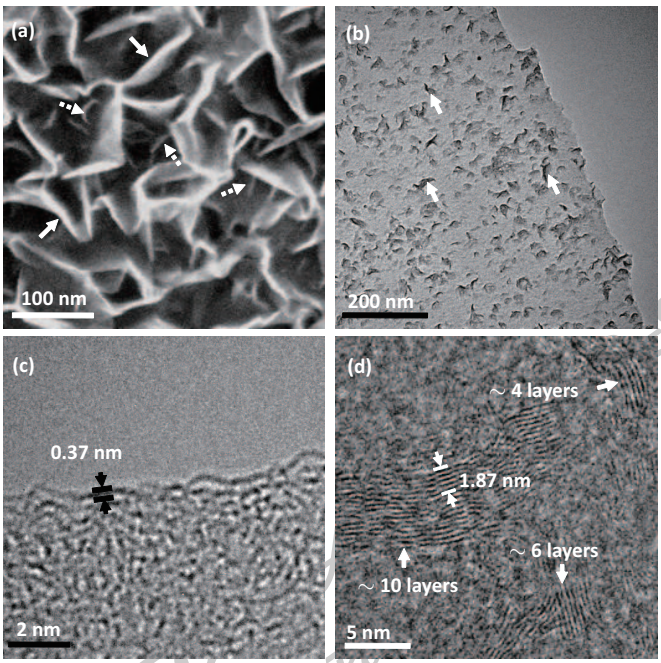


Fig. 2. (a) High-magnification SEM image showing abundant of stem- (solid arrows) and branch- (dashed arrows) structures. (b) TEM image of branches (labeled by arrows) on the stem. (c) HRTEM image at the edge of the stem, indicating the stem is terminated with two-layer graphene with an interlayer spacing of 0.37 nm. (d) HRTEM image at the branches, showing the branches are composed of various graphene layers with an average interlayer spacing of $1.87/5 = 0.374$ nm.

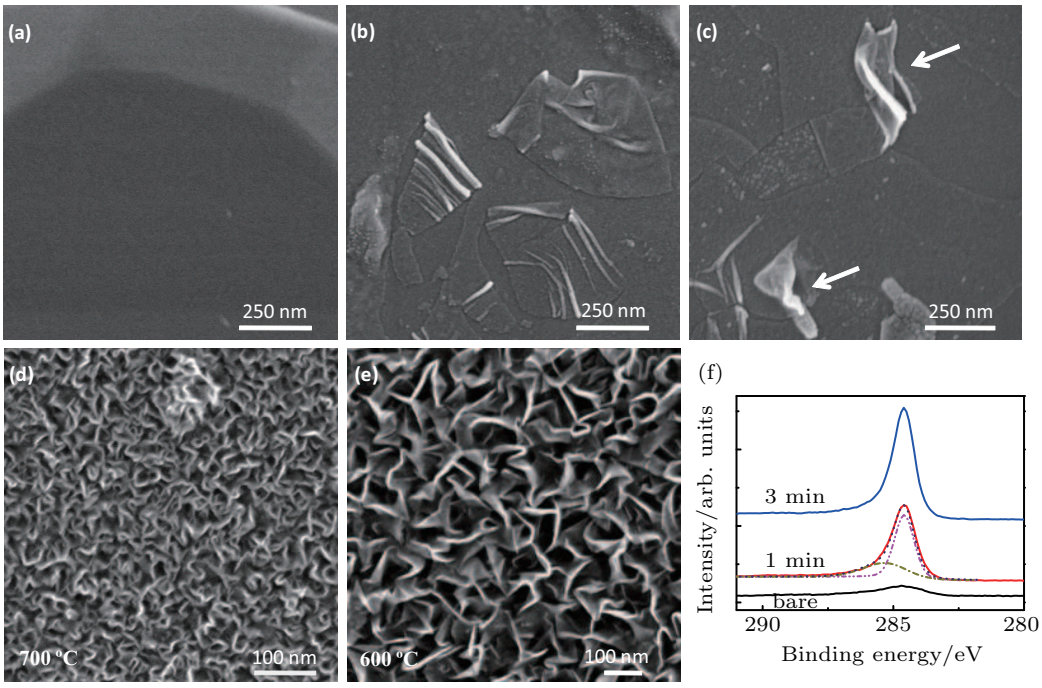


Fig. 3. (a) High-magnification SEM image of bare Al_2O_3 powder. (b) Image after 10-min deposition showing 2D graphene layers. (c) Image after 20-min deposition showing the transition to 3D graphene layers. (d) and (e) Graphene sheets showing variable lateral lengths at different growth temperatures. (f) High-resolution XPS spectra recorded after different deposition times. Graphene deposition parameters are: (b)–(d) 700 °C, 290 W; (e) 600 °C, 290 W.

XPS measurements are also carried out to detect the growth process of GNWs. Before deposition, the bare Al_2O_3 powder is treated at ambient atmosphere at 800 °C for three hours. Our results indicate that the content of residual amorphous carbon significantly decreases from 12.3 at.% to 7.1 at.% after this treatment. The high-resolution C 1s spectra recorded at different deposition times are presented in Fig. 3(f). The amorphous carbon on bare Al_2O_3 powder shows a broad C 1s peak and a maximum at 284.68 eV. The peak recorded after 1-min deposition is obviously different, which is located at 284.79 eV and shows a characteristic shape of C 1s spectrum of graphitic carbon. This peak could be fitted with two peaks at 284.8 eV and 285.5 eV, corresponding to graphitic carbon and C–O bond, respectively. The existence of C–O bond might be due to the fact that the carbon radicals partially reduce the Al_2O_3 surface and the Al–O–C bonds form.[29] There is almost no change in the shape of C 1s when the deposition time is elongated to 3 min. The carbon concentrations measured on the powder surface are 34 at.% and 48 at.% after 1-min and 3-min depositions, respectively.

To elucidate the universality and advantage of our method, we further deposit GNWs on the surface of silicon nitride (Si_3N_4). As shown in the SEM image of Fig. 4(a), the Si_3N_4 powder shows a particle size of below 3 μm. At higher magnification, graphene sheets could be identified with an average length of 70 nm and 150 nm at the growth temperature of 700 °C (Fig. 4(b)) and 600 °C (Fig. 4(c)), respectively. The Raman spectrum in Fig. 4(d) presents three peaks located at 1350 cm^{-1} (D), 1586 cm^{-1} (G), and 2702 cm^{-1} (2D), further verifying the formation of GNWs. The results are similar to these on alumina powder, indicating that the growth of GNWs is independent on the type of ceramic powder and could be further extended to other powders such as SiC, SiO_2, and mullite.

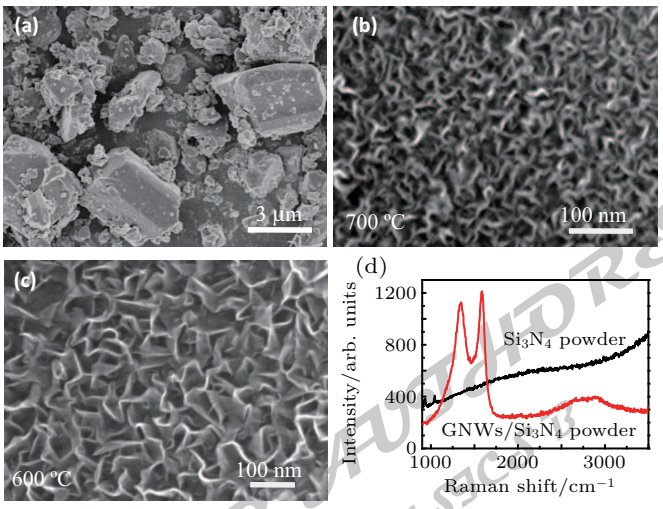


Fig. 4. (a) Low-magnification SEM image showing overall view of Si_3N_4 particles. (b) and (c) High-magnification image of GNWs on the particle, showing variable lateral lengths at different growth temperatures. (d) Raman spectrum. Graphene deposition parameters are: (b) 700 °C, 290 W, 90 min; (c) 600 °C, 290 W, 90 min.

The GNWs-Al_2O_3 powder is sintered by SPS method to fabricate alumina composite. The volume percent of graphene in this sample is about 0.25%. Pure Al_2O_3 powder is also

sintered to bulk monilithic alumina using the same condition for comparison. The composite has a black color, exhibiting a sharp contrast with the white monolithic alumina. Graphene owns exceptional high electrical conductivity and is considered to be an ideal additive to improve this property of ceramic.[7,16,30] We carry out measurements on several physical properties such as bulk density, Vickers hardness, and electrical conductivity. The monilithic alumina and the composite exhibit similar bulk density with the value of 3.899 g/cm^3 and 3.918 g/cm^3, respectively. These numbers are very close to the ideal value of alumina (3.97 g/cm^3), indicating that the sintering parameters used in our experiment are suitable. There are little differences in hardness between the two samples, with the value of 1667 HV and 1652 HV, respectively. However, the addition of graphene could significantly raise the electrical conductivity of alumina. Our experiment results show that the monilithic Al_2O_3 is an insulator. The graphene sheets in the composite provide numerous conductive paths and sharply increase the electrical conductivity to a value of 0.724 S/m.

4. Conclusion and perspectives

GNWs film has been successfully deposited on the surface of Al_2O_3 and Si_3N_4 powders. The layers, counted from HRTEM images, imply that the GNWs correspond to few-layer graphene. SEM results indicate that the growth process includes a transition from 2D to 3D mode at the initial stage. The lateral length of graphene sheets could be well controlled by changing the deposition temperature, which is about 70 nm and 150 nm at 700 °C and 600 °C, respectively. Moreover, the electrical conductivity of the composite significantly increases compared with the monilithic alumina. Our experiment provides an alternative way to realize the uniform mixing of graphene and ceramic powders, which has potential applications in fabricating ceramic matrix composites.

References

[1] Inam F, Yan H X, Jayaseelan D D, Peijs T and Reece M J 2010 *J. Eur. Ceram. Soc.* **30** 153

[2] Cho J, Boccaccini A R and Shaffer M S P 2009 *J. Mater. Sci.* **44** 1934

[3] Cho J, Inam F, Reece M J, Chlup Z, Dlouhy I, Shaffer M S P and Boccaccini A R 2011 *J. Mater. Sci.* **46** 4770

[4] Wu Y and Kim G Y 2011 *J. Mater. Process. Technol.* **211** 1341

[5] Esawi A and Morsi K 2007 *Compos. Part. A* **38** 646

[6] Esawi A M K, Morsi K, Sayed A, Taher M and Lanka S 2011 *Compos. Part. A* **42** 234

[7] Fan Y C, Wang L J, Li J L, Li J Q, Sun S K, Chen F, Chen L D and Jiang W 2010 *Carbon* **48** 1743

[8] He T, Li J L, Wang L J, Zhu J J and Jiang W 2009 *Mater. Trans.* **50** 749

[9] Tapaszto O, Tapaszto L, Marko M, Kern F, Gadow R and Balazsi C 2011 *Chem. Phys. Lett.* **511** 340

[10] Liu J, Yan H X and Jiang K 2013 *Ceram. Int.* **39** 6215

[11] Cygan T, Wozniak J, Kostecki M, Petrus M, Jastrzębska A, Ziemkowska W and Olszyna A 2017 *Ceram. Int.* **43** 6180

[12] Sedlák R, Kovalčíková A, Múdra E, Rutkowski P, Dubiel A, Girman V, Bystrický R and Duszaa J 2017 *J. Eur. Ceram. Soc.* **37** 3773

[13] Walker L S, Marotto V R, Rafiee M A, Koratkar N and Corral E L 2011 *ACS Nano* **5** 3182

[14] Wang K, Wang Y F, Fan Z J, Yan J and Wei T 2011 *Mater. Res. Bull.* **46** 315

[15] Gutierrez-Gonzalez C F, Smirnov A, Centeno A, Fernández A, Alonso B, Rocha V G, Torrecillas R, Zurutuza A and Bartolome J F 2015 *Ceram. Int.* **41** 7434

[16] Porwal H, Grasso S and Reece M J 2013 *Adv. Appl. Ceram.* **112** 443

[17] Markandana K, Chin J K and Tan M 2017 *J. Mater. Res.* **32** 84

[18] Zhang Y, Du J L, Tang S, Liu P, Deng S Z, Chen J and Xu N S 2012 *Nanotechnol.* **23** 015202

[19] Jiang L L, Yang T Z, Liu F, Dong J, Yao Z H, Shen C M, Deng S Z, Xu N S, Liu Y Q and Gao H J 2013 *Adv. Mater.* **25** 250

[20] Sun J Y, Chen Y B, Cai X, Ma B J, Chen Z L, Priydarshi M K, Chen K, Gao T, Song X J, Ji Q Q, Guo X F, Zou D C, Zhang Y F and Liu Z F 2015 *Nano Res.* **8** 3496

[21] Chen X D, Chen Z L, Sun J Y, Zhang Y F and Liu Z F 2016 *Acta Phys. -Chim. Sin.* **32** 14

[22] Zhou H T, Yu N, Zou F, Yao Z H, Gao G and Shen C M 2016 *Chin. Phys. B* **25** 096106

[23] Zhou H T, Liu D B, Luo F, Luo B W, Tian Y, Chen D S and Shen C M 2018 *Micro & Nano Lett.* **13** 842

[24] Zou F, Zhou H T, Yu N, Yao Z H, Liu F and Shen C M 2016 *Chem. Phys. Lett.* **664** 29

[25] Dong J, Yao Z H, Yang T Z, Jiang L L and Shen C M 2013 *Sci. Rep.* **3** 1733

[26] Gupta A, Chen G, Joshi P, Tadigadapa S and Eklund P C 2006 *Nano Lett.* **6** 2667

[27] Graf D, Molitor F, Ensslin K, Stampfer C, Jungen A, Hierold C and Wirtz L 2007 *Nano Lett.* **7** 238

[28] Vitchev R, Alexander Malesevic A, Petrov R, Kemps R, Mertens M, Annick Vanhulsel A and Haesendonck C 2010 *Nanotechnol.* **21** 095602

[29] Zhou M, Bi H, Lin T Q, Lv X J, Wan D Y, Huang F Q and Lin J H 2014 *Carbon* **75** 314

[30] Li Q S, Zhang Y J, Gong H Y, Sun H B, Li T, Guo X and Ai S H 2015 *Ceram. Int.* **41** 13547

Chinese Physics B

Volume 28 Number 6 June 2019

(*Continued on the Bookbinding Inside Back Cover*)

ATOMIC AND MOLECULAR PHYSICS

ELECTROMAGNETISM, OPTICS, ACOUSTICS, HEAT TRANSFER, CLASSICAL MECHANICS, AND FLUID DYNAMICS